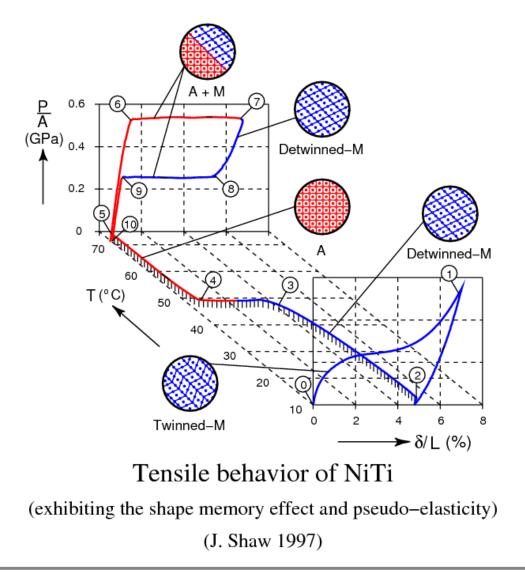
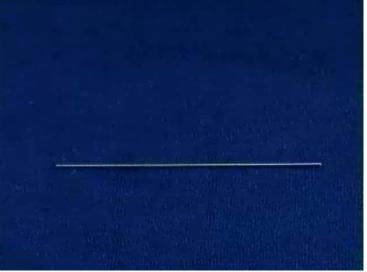




#### PHASE TRANSFORMATIONS IN SHAPE MEMORY ALLOYS







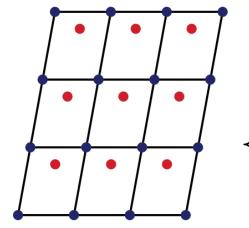


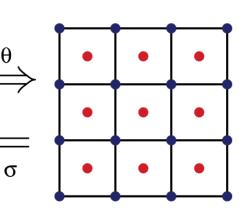


#### PHASE TRANSFORMATIONS: LATTICE INSTABILITIES









θ

(lower symmetry)

(higher symmetry)

- Shape memory behavior due to instabilities of the atomic lattice:
- At higher temperatures: austenitic (higher symmetry phase) is stable.
- At higher stresses: martensitic (lower symmetry phase) is stable.

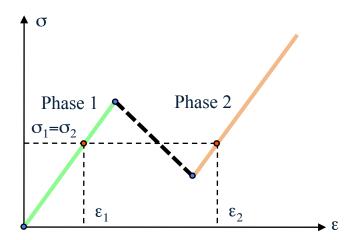
 The consequence of this lattice-level instability shows all the way up to structural scale.

## PHASE TRANSFORMATIONS: CONTINUUM

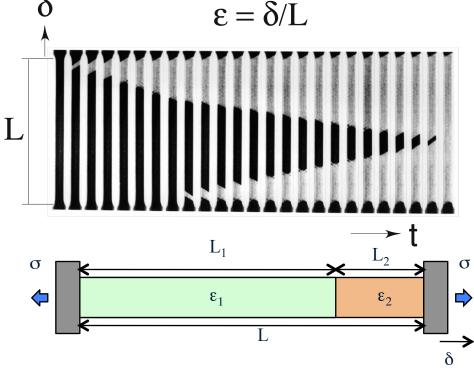
#### **STRESS-INDUCED P.T. IN THIN STRIP STRECTCHING**



COEXISTING DIFFERENT PHASES IN CuAINi (Chu & James 1995)



PROPAGATING INSTABILITY IN A NITI STRIP (J. SHAW, PhD 1997)  $\delta = \frac{\delta}{J}$ 



• Stress is not a monotonically increasing function of strain, due to the presence of two different phases.

• The solution to the uniaxial stretching strip problem has discontinuities and is not unique.

**PHASE TRANSFORMATIONS: CONTINUUM** 

#### **DISCONTINUOUS EQUILIBRIA IN THIN STRIP STRECTCHING**

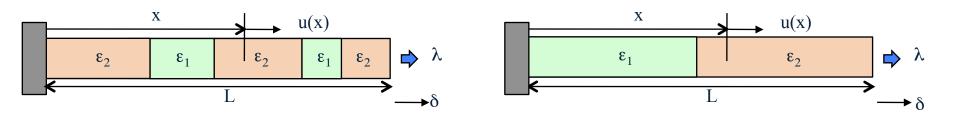
$$\mathcal{E}(u,\lambda) = \int_{0}^{L} W(\varepsilon) dx - \lambda u(L); \text{ energy stored in bar where } : W(\varepsilon) \equiv \int_{0}^{\varepsilon} \sigma(\varepsilon) \delta\epsilon$$

$$\varepsilon(x) = \frac{du}{dx}; \ u(0) = 0, \ u(L) = \delta; \text{ kinematics and b.c.}$$

$$\mathcal{E}_{,u} \, \delta u = \int_{0}^{L} \left[ \frac{dW(\varepsilon)}{d\varepsilon} \delta \varepsilon \right] dx - \lambda \delta u(L) = 0; \text{ equilibrium,}$$

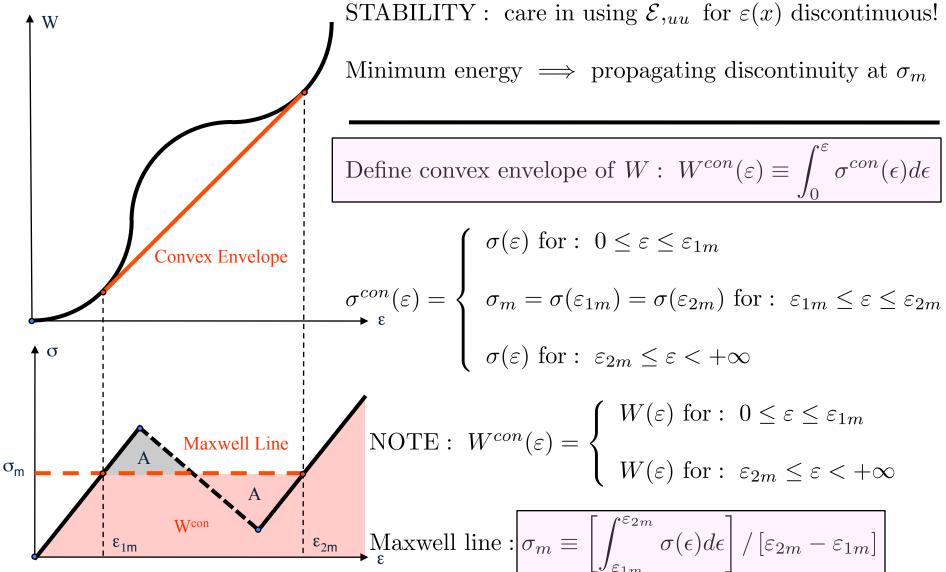
$$\Longrightarrow \frac{dW(\varepsilon)}{d\varepsilon} \equiv \sigma(\varepsilon(x)) = \lambda; \text{ bar has constant stress } \lambda,$$

Piecewise contant strain solution :  $\sigma(\varepsilon_1) = \sigma(\varepsilon_2) = \lambda$ . SOLUTION IS NOT UNIQUE!



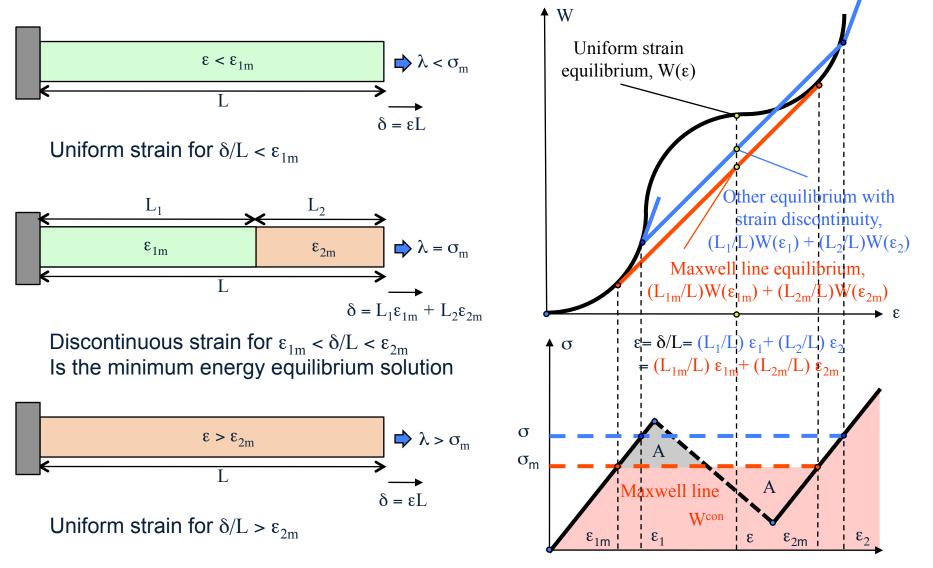
# PHASE TRANSFORMATIONS: CONTINUUM

#### CONVEXIFIED ENERGY AND THE MAXWELL LINE SOLUTION





#### STABILITY OF THE MAXWELL LINE EQUILIBRIUM SOLUTION







#### **TEMPERATURE-INDUCED P.T.: LATTICE MODELS**



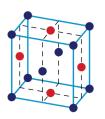
**B2 - AUSTENITE FOR:** NiTi, AuCd, CuAlNi, NiAl

L1<sub>0</sub> - MARTENSITE FOR:

NiAl

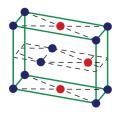
NiTi

- Some common SMA crystals have bi-atomic lattices.
- Their Austenitic (high symmetry) phase is cubic.

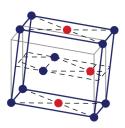


 Their Martensitic (low symmetry) phase can be tetragonal (e.g. L1<sub>0</sub> for NiAl), orthorhombic (e.g. B19 for AuCd, CuAlNi) or monoclinic (e.g. B19' for NiTi).

 Their shape memory effect will be modeled by using temperature-dependent atomic potentials.



 Using Cauchy-Born hypothesis, we derive the **B19 - MARTENSITE FOR:** continuum energy density of the infinite, perfect, AuCd, CuAlNi crystal  $W(U,P,\theta)$ , where U is the macroscopic stretch tensor, **P** shifts and  $\theta$  is a temperature parameter.



 Martensitic phase transformations are identified as bifurcated equilibrium solutions emerging from the B19' - MARTENSITE FOR: principal (austenitic) equilibrium path.

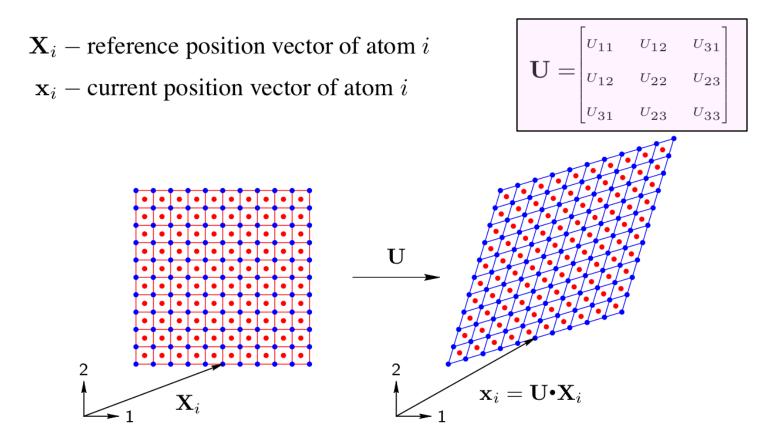
*JMPS*, Elliott et al., **50**, 2002, p.2463; *JJSS*, Elliott et al., **39**, 2002, p.3845; *JMPS*, Elliott et al. I & II, **54**, 2006, p.161 & p.193





#### **CAUCHY HYPOTHESIS – UNIFORM DEFORMATION**

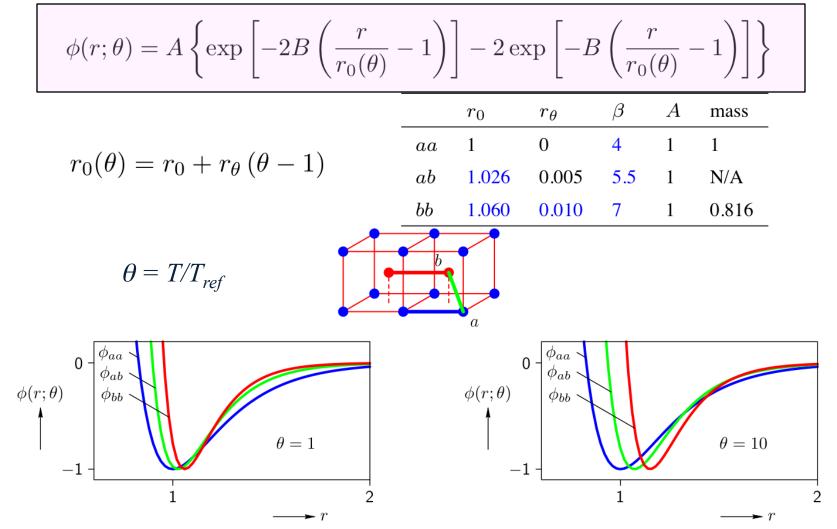
• Continuum-Lattice connection:  $\mathbf{U}$  – symmetric right stretch tensor



• Atomic separation  $r_{ij} = \|\mathbf{x}_i - \mathbf{x}_j\| = \sqrt{(\mathbf{X}_i - \mathbf{X}_j) \cdot \mathbf{U}^2 \cdot (\mathbf{X}_i - \mathbf{X}_j)}$ 



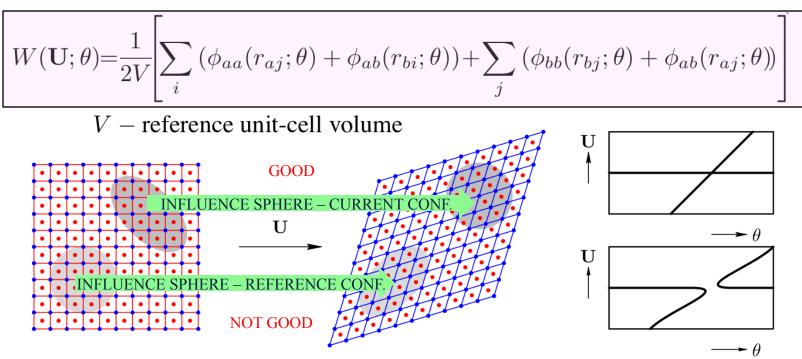
#### **TEMPERATURE-DEPENDENT PAIR POTENTIALS**



**NOTE:** Method easily generalized to much more general interaction potentials







• Determine *stress-free* equilibrium solutions: (solve for 6 components of U)

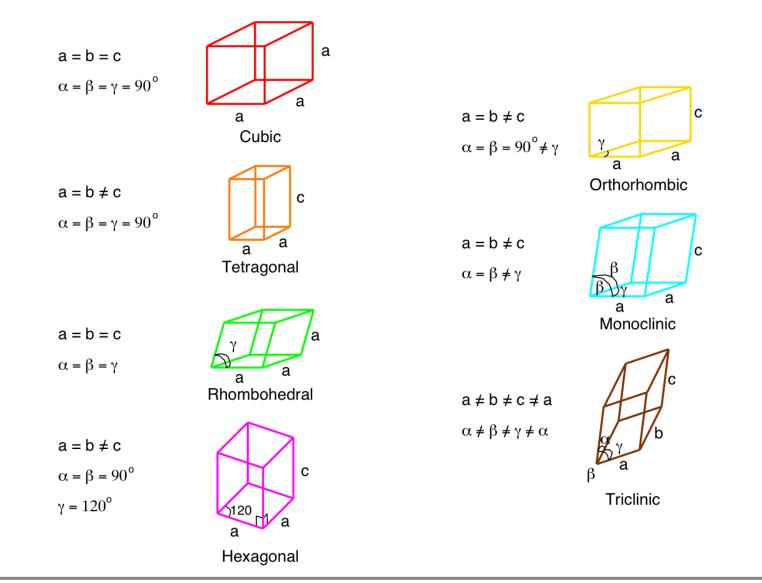
$$\frac{\partial W(\mathbf{U};\theta)}{\partial \mathbf{U}} = \mathbf{0},$$

• Stability (∞ wavelength):

$$\frac{\partial^2 W}{\partial \mathbf{U} \partial \mathbf{U}}$$
 is positive definite.

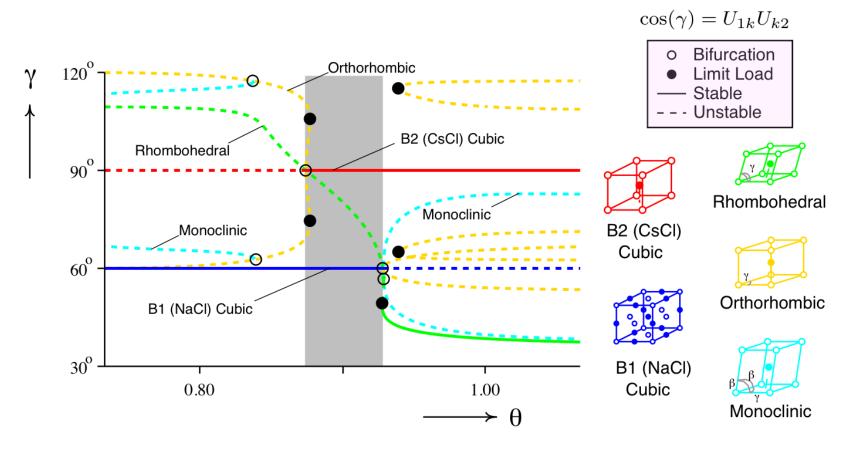


#### THE SEVEN CRYSTAL SYSTEMS





#### **CAUCHY HYPOTHESIS – FINAL RESULTS**

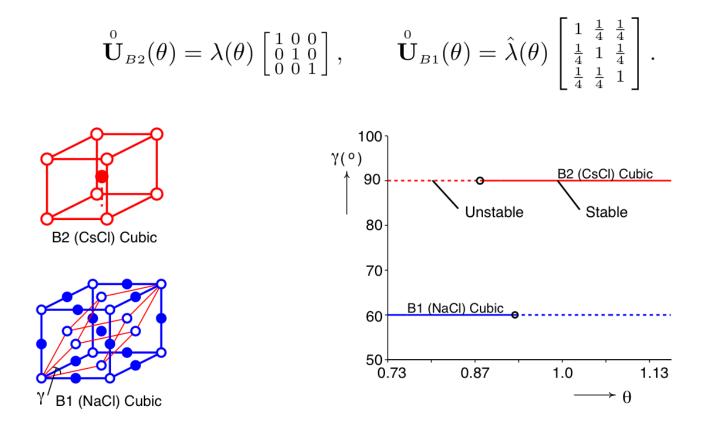


- Overlapping stable segments of B2 (CsCl) and B1 (NaCl) cubic phases
- Multiple bifurcation points require asymptotic techniques



#### **CAUCHY HYPOTHESIS – PRINCIPAL SOLUTIONS (CUBIC)**

Stress-Free Equilibrium Paths of Cubic Phases



NOTE: Two different cubic symmetry solutions exist. From the BCC B2 crystal one can obtain a FCC B1 crystal by applying a particular stretch tensor U.

## PHASE TRANSFORMATIONS: ATOMISTIC



#### **CAUCHY HYPOTHESIS – MODULI OF CUBIC SOLUTIONS**

• Cubic material with principal solution  $\overset{_{0}}{\mathbf{U}}(\theta) = \lambda(\theta)\mathbf{I}$ .

 $W(\mathbf{U};\boldsymbol{\theta}) = W(\mathbf{M}^T \boldsymbol{\cdot} \mathbf{U} \boldsymbol{\cdot} \mathbf{M};\boldsymbol{\theta})$ 

where  $\mathbf{M}$  is in the octahedral symmetry group.

Thus,

$$W = W(U_{ii}, U_{jj}, U_{kk}, U_{ij}, U_{jk}, U_{ki}; \theta)$$
  
=  $W(U_{ii}, U_{jj}, U_{kk}, -U_{ij}, -U_{jk}, U_{ki}; \theta)$   
=  $W(U_{ii}, U_{jj}, U_{kk}, -U_{ij}, U_{jk}, -U_{ki}; \theta)$   
=  $W(U_{ii}, U_{jj}, U_{kk}, U_{ij}, -U_{jk}, -U_{ki}; \theta)$  (no sum)

**NOTE:** For stability calculations most of the components of the energy's second derivatives  $\partial^2 W / \partial U_{ij} \partial U_{kl}$  are related due to the cubic symmetry of the principal solution. Here M are the orthonormal matrices of the full octahedral symmetry group.





#### **CAUCHY HYPOTHESIS – MODULI OF CUBIC SOLUTIONS**

$$W(\overset{0}{\mathbf{U}}(\theta) + \mathbf{\Delta}\mathbf{U}; \theta) = \overset{0}{L} + \overset{1}{L}_{n} \sum_{i} \Delta U_{ii}$$

$$+ \frac{1}{2!} \left( \overset{2}{L}_{nn} \sum_{i} (\Delta U_{ii})^{2} + \overset{2}{L}_{nn'} \sum_{i \neq j} (\Delta U_{ii} \Delta U_{jj}) + \overset{2}{L}_{ss} \sum_{i \neq j} 2(\Delta U_{ij})^{2} \right)$$

$$+ \frac{1}{3!} \left( \overset{3}{L}_{nnn} \sum_{i} (\Delta U_{ii})^{3} + \overset{3}{L}_{nnn'} \sum_{i \neq j} 3((\Delta U_{ii})^{2} \Delta U_{jj}) + \overset{3}{L}_{nn'n''} \sum_{i \neq j \neq k \neq i} (\Delta U_{ii} \Delta U_{jj} \Delta U_{kk}) + \overset{3}{L}_{nss} \sum_{i \neq j} 12(\Delta U_{ii} (\Delta U_{ij})^{2})$$

$$+ \overset{3}{L}_{n'ss} \sum_{i \neq j \neq k \neq i} 6(\Delta U_{ii} (\Delta U_{jk})^{2}) + \overset{3}{L}_{ss's''} \sum_{i \neq j \neq k \neq i} 8(\Delta U_{ij} \Delta U_{ik} \Delta U_{jk}) \right)$$

$$+ O(\mathbf{A} \mathbf{U}^{4}) \qquad \text{NOTE: Only 3 independent components of moduli 1}$$

 $+Oig({f \Delta U}^4ig)$  . NOTE: Only 3 independent components of moduli !

$$\overset{2}{L}_{nn} = \frac{\partial^2 W}{\partial U_{11} \partial U_{11}}, \qquad \overset{2}{L}_{nn'} = \frac{\partial^2 W}{\partial U_{11} \partial U_{22}}, \qquad \overset{2}{L}_{ss} = \frac{\partial^2 W}{\partial U_{12} \partial U_{12}}$$





#### **CAUCHY HYPOTHESIS – CRITICAL POINTS ON B2, B1**

- Identify critical points,  $\frac{\partial^2 W}{\partial \mathbf{U} \partial \mathbf{U}} \Big|_c \mathbf{U}^{(I)} = 0.$   $I = 1, \dots, H$
- $\stackrel{(I)}{\mathbf{U}}$  is the  $I^{\text{th}}$  eigenvector of  $\frac{\partial^2 W}{\partial \mathbf{U} \partial \mathbf{U}}\Big|_c$ , with eigenvalue 0.

$$\begin{bmatrix} \overset{2}{L_{nn}^{c}} & \overset{2}{L_{nn'}^{c}} & \overset{2}{L_{nn'}^{c}} & 0 & 0 & 0\\ \overset{2}{L_{nn'}^{c}} & \overset{2}{L_{nn}^{c}} & \overset{2}{L_{nn'}^{c}} & 0 & 0 & 0\\ \overset{2}{L_{nn'}^{c}} & \overset{2}{L_{nn'}^{c}} & \overset{2}{L_{nn'}^{c}} & 0 & 0 & 0\\ 0 & 0 & 0 & \overset{2}{L_{ss}^{c}} & 0 & 0\\ 0 & 0 & 0 & 0 & \overset{2}{L_{ss}^{c}} & 0 & 0\\ 0 & 0 & 0 & 0 & \overset{2}{L_{ss}^{c}} & 0\\ 0 & 0 & 0 & 0 & 0 & \overset{2}{L_{ss}^{c}} \end{bmatrix} \begin{bmatrix} U_{11} \\ U_{22} \\ U_{33} \\ 2U_{12} \\ 2U_{23} \\ 2U_{31} \end{bmatrix} = \begin{bmatrix} 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \end{bmatrix}$$

• Thus,

$$\det\left(\left.\frac{\partial^2 W}{\partial \mathbf{U} \partial \mathbf{U}}\right|_c\right) = \left(L_{nn}^c + 2L_{nn'}^c\right) \left(L_{nn}^c - L_{nn'}^c\right)^2 \left(L_{ss}^c\right)^3 = 0.$$





• Expansion of U and  $\theta$  near  $\theta_c$ .

 $(\xi - bifurcation amplitude)$ 

 $(\stackrel{(I)}{\mathbf{U}}$  - eigenvectors in the null-space of  $W^{c},_{\mathbf{U}\mathbf{U}}$ )

$$\mathbf{U}(\xi) = \overset{0}{\mathbf{U}}(\theta(\xi)) + \xi \left(\sum_{I=1}^{H} \alpha_{I} \overset{(I)}{\mathbf{U}}\right) + O(\xi^{2}); \quad \theta(\xi) = \theta_{c} + \xi \theta_{1} + \frac{\xi^{2}}{2} \theta_{2} + O(\xi^{3}).$$

• Equilibrium equations:  $\alpha$  – tangent vector,  $\mathcal{E}$  – higher order derivatives of W

if 
$$\theta_1 \neq 0$$
 (asymmetric):  $\sum_{J,K=1}^{H} \alpha_J \alpha_K \mathcal{E}_{IJK} + 2\theta_1 \sum_{J=1}^{H} \alpha_J \mathcal{E}_{IJ\theta} = 0,$ 

or

if 
$$\theta_1 = 0$$
 (symmetric):  $\sum_{J,K,L=1}^{H} \alpha_J \alpha_K \alpha_L \mathcal{E}_{IJKL} + 3\theta_2 \sum_{J=1}^{H} \alpha_J \mathcal{E}_{IJ\theta} = 0.$ 





• Initial stability depends on the eigenvalues of  $B_{IJ}$ .

$$B_{IJ} = \begin{cases} \theta_1 \mathcal{E}_{IJ\theta} + \sum_{K=1}^{H} \alpha_K \mathcal{E}_{IJK} & \text{if } \theta_1 \neq 0 \text{ (asymmetric),} \\ \theta_2 \mathcal{E}_{IJ\theta} + \sum_{K,L=1}^{H} \alpha_K \alpha_L \mathcal{E}_{IJKL} & \text{if } \theta_1 = 0 \text{ (symmetric).} \end{cases}$$

$$\mathcal{E}_{IJK} \equiv \left. \frac{\partial^3 W}{\partial U_{ij} \partial U_{kl} \partial U_{mn}} \right|_c {}^{(I)} {}^{(J)}_{Uij} {}^{(K)}_{Ukl} {}^{(K)}_{mn},$$

$$\mathcal{E}_{IJKL} \equiv \left( \left( \left( \frac{\partial^4 W}{\partial U_{ij} \partial U_{kl} \partial U_{mn} \partial U_{qr}} \Big|_c \stackrel{(J)}{U_{kl}} \stackrel{(J)}{U_{mn}} \stackrel{(L)}{U_{qr}} + \dots \right) \stackrel{(I)}{U_{ij}} \right)$$

$$\mathcal{E}_{IJ\theta} \equiv \left. \left( \frac{d}{d\theta} \left( \frac{\partial^2 W(\overset{0}{U}(\theta); \theta)}{\partial U_{ij} \partial U_{kl}} \right) \right) \right|_c^{(I)} \overset{(J)}{U_{ij} U_{kl}}.$$



$$\left( \frac{\partial^2 W}{\partial \mathbf{U} \partial \theta} \Big|_c \right) \mathbf{U} = \begin{cases} = 0 & \text{bifurcation} \\ \neq 0 & \text{limit load} \end{cases} \quad I = 1, \dots, H$$

• Case I. 
$$\begin{bmatrix} 2 \\ L_{nn}^c + 2L_{nn'}^c = 0 \end{bmatrix}$$
, multiplicity of  $H = 1$ 

$$\mathbf{\ddot{U}} = \begin{bmatrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & 1 \end{bmatrix}$$

$$\left( \left. \frac{\partial^2 W}{\partial \mathbf{U} \partial \theta} \right|_c \right)^{(1)} = 3 \left. \frac{\partial \tilde{L}_n}{\partial \theta} \right|_c \neq 0$$
$$\implies \text{Limit Load.}$$



• Case II. 
$$L_{nn}^{2} = L_{nn'}^{2}$$
, multiplicity of  $H = 2$ 

$$\overset{(1)}{\mathbf{U}} = \begin{bmatrix} 2 & 0 & 0 \\ 0 & -1 & 0 \\ 0 & 0 & -1 \end{bmatrix}, \quad \overset{(2)}{\mathbf{U}} = \begin{bmatrix} -1 & 0 & 0 \\ 0 & 2 & 0 \\ 0 & 0 & -1 \end{bmatrix}.$$

$$\left(\left.\frac{\partial^2 W}{\partial \mathbf{U} \partial \theta}\right|_c\right) \mathbf{\dot{U}}^{(I)} = \left.\frac{\partial \hat{L}_n}{\partial \theta}\right|_c \left(\sum_{i=1}^3 \overset{(I)}{U_{ii}}\right) = 0$$

 $\implies$  Bifurcation.



Case II. 
$$\begin{bmatrix} 2 \\ L_{nn}^c = L_{nn'}^c \end{bmatrix}$$

 $H = 2 \Rightarrow$  a max of  $2^{H} - 1 = 3$ , asymmetric ( $\theta_{1} \neq 0$ ), bifurcated paths,

$$\mathbf{U} = \begin{bmatrix} c & 0 & 0 \\ 0 & a & 0 \\ 0 & 0 & a \end{bmatrix}, \qquad \begin{bmatrix} a & 0 & 0 \\ 0 & c & 0 \\ 0 & 0 & a \end{bmatrix}, \qquad \begin{bmatrix} a & 0 & 0 \\ 0 & a & 0 \\ 0 & 0 & c \end{bmatrix}.$$
$$\mathbf{\alpha} = (1,0) \qquad (0,1) \qquad (-1,-1)$$

All 3 bifurcated paths are *initially* unstable.



• Case III. 
$$L_{ss}^2 = 0$$
, multiplicity  $H = 3$ 

$$\stackrel{(1)}{\mathbf{U}} = \begin{bmatrix} 0 & 0 & 0 \\ 0 & 0 & 1 \\ 0 & 1 & 0 \end{bmatrix}, \quad \stackrel{(2)}{\mathbf{U}} = \begin{bmatrix} 0 & 0 & 1 \\ 0 & 0 & 0 \\ 1 & 0 & 0 \end{bmatrix}, \quad \stackrel{(3)}{\mathbf{U}} = \begin{bmatrix} 0 & 1 & 0 \\ 1 & 0 & 0 \\ 0 & 0 & 0 \end{bmatrix}.$$

$$\left(\left.\frac{\partial^2 W}{\partial \mathbf{U} \partial \theta}\right|_c\right) \mathbf{\dot{U}}^{(I)} = \left.\frac{\partial \hat{L}_n}{\partial \theta}\right|_c \left(\sum_{i=1}^3 U_{ii}^{(I)}\right) = 0$$

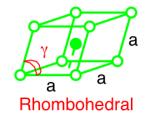
 $\implies$  Bifurcation.





Case III.  $\begin{bmatrix} 2 \\ L_{ss}^c = 0 \end{bmatrix}$ ,  $H = 3 \Rightarrow a \max \text{ of } 2^H - 1 = 7 \text{ bifurcated paths.}$ 

**a**) Four, asymmetric ( $\theta_1 \neq 0$ ), bifurcated paths,



$$\mathbf{U} = \begin{bmatrix} a & \zeta & \zeta \\ \zeta & a & \zeta \\ \zeta & \zeta & a \end{bmatrix}, \qquad \begin{bmatrix} a & \zeta & \zeta \\ \zeta & a & -\zeta \\ \zeta & -\zeta & a \end{bmatrix}, \qquad \begin{bmatrix} a & \zeta & -\zeta \\ \zeta & a & \zeta \\ -\zeta & \zeta & a \end{bmatrix}, \qquad \begin{bmatrix} a & -\zeta & \zeta \\ -\zeta & a & \zeta \\ \zeta & \zeta & a \end{bmatrix}.$$
$$\boldsymbol{\alpha} = (-1, -1, -1) / \sqrt{3} \qquad (-1, 1, 1) / \sqrt{3} \qquad (1, -1, 1) / \sqrt{3} \qquad (1, 1, -1) / \sqrt{3}$$

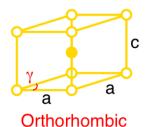
All 4 rhombohedral bifurcated paths are *initially* unstable.





Case III.  $\begin{bmatrix} 2 \\ L_{ss}^c = 0 \end{bmatrix}$ ,  $H = 3 \Rightarrow$  a max of  $2^H - 1 = 7$  bifurcated paths.

**b**) Three, symmetric ( $\theta_1 = 0; \theta_2 \neq 0$ ), bifurcated paths,



$$\mathbf{U} = \begin{bmatrix} c & 0 & 0 \\ 0 & a & \zeta \\ 0 & \zeta & a \end{bmatrix}, \qquad \begin{bmatrix} a & 0 & \zeta \\ 0 & c & 0 \\ \zeta & 0 & a \end{bmatrix}, \qquad \begin{bmatrix} a & \zeta & 0 \\ \zeta & a & 0 \\ 0 & 0 & c \end{bmatrix}.$$
$$\boldsymbol{\alpha} = (1, 0, 0) \qquad (0, 1, 0) \qquad (0, 0, 1)$$

All 3 orthorhombic bifurcated paths' *initial* stability depends on W.

PHASE TRANSFORMATIONS: ATOMISTIC



Procedure

- Cubic Phase
  - $U_{11} = U_{22} = U_{33} = a,$  $U_{12} = U_{23} = U_{31} = 0.$
- Rhombohedral Phase

$$U_{11} = U_{22} = U_{33} = a,$$

$$U_{12} = U_{23} = U_{31} = \zeta.$$

• Orthorhombic Phase

$$U_{33} = c, \quad U_{11} = U_{22} = a,$$
  
 $U_{12} = \zeta, \quad U_{23} = U_{31} = 0.$ 

• Monoclinic Phase

$$U_{33} = c, \quad U_{11} = U_{22} = a,$$
  
 $U_{12} = \zeta, \quad U_{23} = U_{31} = \rho.$ 

Solve: 
$$\frac{\partial W}{\partial a} = 0.$$

Solve: 
$$\frac{\partial W}{\partial a} = 0, \ \frac{\partial W}{\partial \zeta} = 0.$$

Solve: 
$$\frac{\partial W}{\partial a} = 0, \ \frac{\partial W}{\partial c} = 0, \ \frac{\partial W}{\partial \zeta} = 0.$$

Solve: 
$$\begin{aligned} \frac{\partial W}{\partial a} &= 0, \qquad \frac{\partial W}{\partial \zeta} &= 0, \\ \frac{\partial W}{\partial c} &= 0, \qquad \frac{\partial W}{\partial \rho} &= 0. \end{aligned}$$



### CAUCHY HYPOTHESIS – FOLLOWING EQUILIBRIUM PATHS

**PHASE TRANSFORMATIONS: ATOMISTIC** 

• For each path prescribe the appropriate amplitude parameter  $\xi$ ,

Cubic $\rightarrow$  $\xi = U_{11} = a,$ Rhombohedral $\rightarrow$  $\xi = U_{12} = \zeta,$ Orthorhombic $\rightarrow$  $\xi = U_{12} = \zeta,$ Monoclinic $\rightarrow$  $\xi = U_{31} = \rho.$ 

- Incremental Newton-Raphson method is used to find solutions.
- Arc-length ( $\Delta d$ ) technique is used to accommodate limit loads.

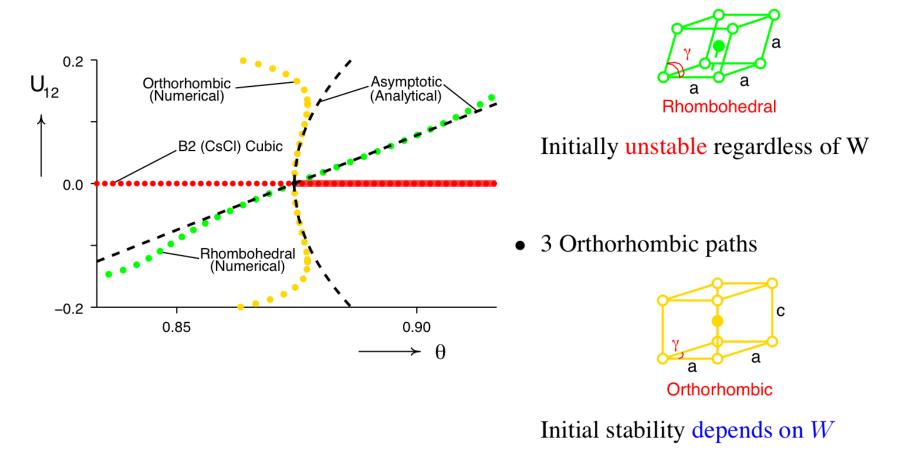
Cubic	$\rightarrow$	$(\Delta d)^2 = (\Delta \theta / \theta^*)^2 + (\Delta a)^2,$
Rhombohedral	$\rightarrow$	$(\Delta d)^2 = (\Delta \theta / \theta^*)^2 + (\Delta a)^2 + (\Delta \zeta)^2,$
Orthorhombic	$\rightarrow$	$(\Delta d)^2 = (\Delta \theta / \theta^*)^2 + (\Delta a)^2 + (\Delta \zeta)^2 + (\Delta c)^2,$
Monoclinic	$\rightarrow$	$(\Delta d)^2 = (\Delta \theta/\theta^*)^2 + (\Delta a)^2 + (\Delta \zeta)^2 + (\Delta c)^2 + (\Delta \rho)^2.$



#### **CAUCHY HYPOTHESIS – VERIFICATION AT TRIPLE POINT**

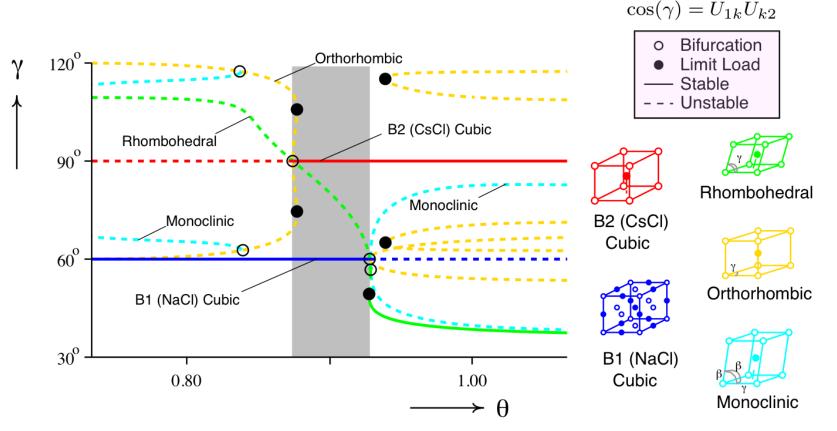
Shear modulus  $L_{ss}^2 = 0$ 

• 4 Rhombohedral paths





#### CAUCHY HYPOTHESIS – REVIEW OF FINAL RESULTS



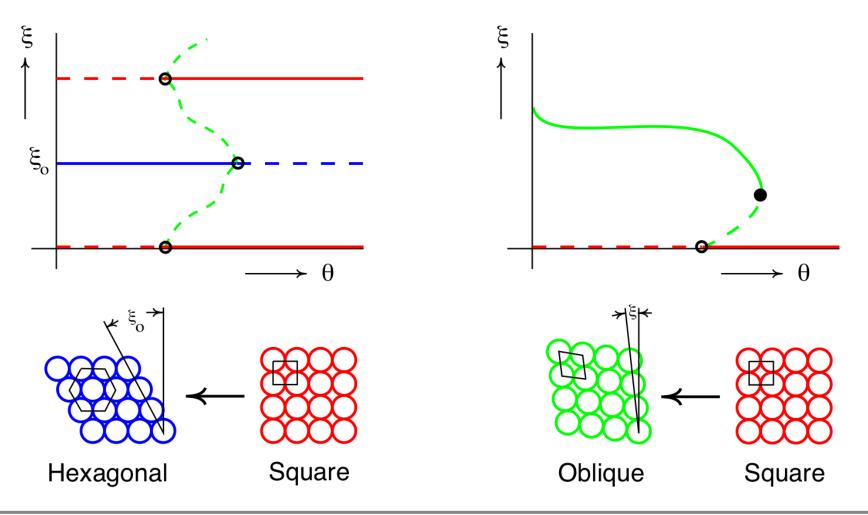
• NOTE: Cauchy hypothesis predicts that only cubic configurations are stable and hence only reconstructive phase transformations found. Model is inadequate for proper M. T. s.

• NOTE: Stability results based on continuum energy minimization are incomplete, since they ignore perturbations with wavelengths of the order of interatomic distances.



#### **CAUCHY HYPOTHESIS – REVIEW OF FINAL RESULTS**

Reconstructive M.T. (No group-subgroup relationship) Proper M.T. (group-subgroup relationship)





#### **PHASE TRANSFORMATIONS: ATOMISTIC**

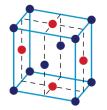


#### **CAUCHY HYPOTHESIS – HOW TO IMPROVE**



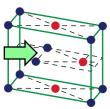
**B2 - AUSTENITE FOR:** NiTi, AuCd, CuAlNi, NiAl

 Simple Cauchy hypothesis predicts stable B1 and **B2** equilibrium paths at given temperature but not the martensitic phases for AuCd, CuAlNi or NiTi.

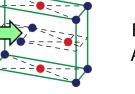


L1<sub>0</sub> - MARTENSITE FOR: NiAl

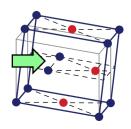
 Reason: Proper martensitic transformations of interest (B19, B19') involve internal shifts. Consequently models with more d.o.f. are needed to describe the equilibrium paths (Cauchy-Born hypothesis) which allow internal shifts in addition to a macroscopic stretch tensor U.



**B19 - MARTENSITE FOR:** AuCd, CuAlNi



 Stability calculations need to consider bounded perturbations of all possible wavelengths with respect to interatomic distances. Consequently phonon spectra calculations (the discrete analogue of the Bloch wave calculations for the continuum periodic solids of the previous lecture) need to be considered.

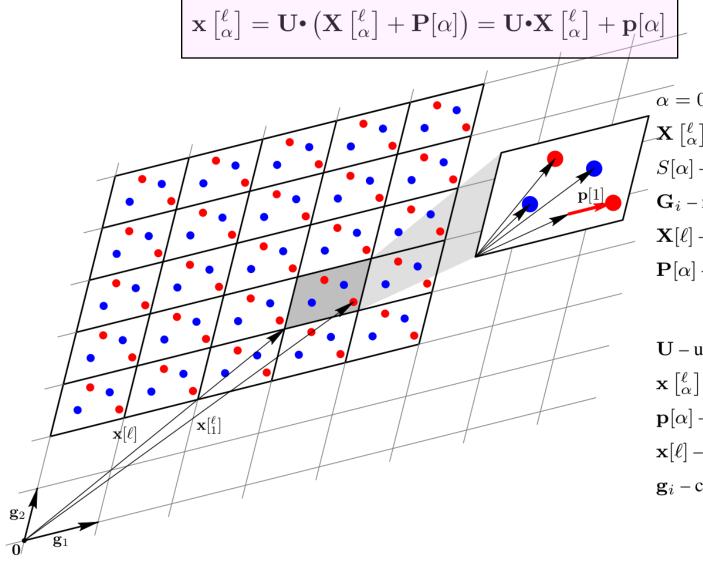


**B19' - MARTENSITE FOR:** NiTi

 Phonon spectra calculations are used to determine the minimum number of d.o.f. needed for the Cauchy-Born calculations.



#### **CAUCHY-BORN HYPOTHESIS (INTERNAL SHIFTS)**



 $\alpha = 0, 1, 2, 3$   $\mathbf{X} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} - \text{reference pos.}$   $S[\alpha] - \text{fractional pos.}$   $\mathbf{G}_i - \text{ref. lattice basis}$   $\mathbf{X}[\ell] - \text{unit-cell ref. pos.}$  $\mathbf{P}[\alpha] - \text{sub-lat. ref. shifts}$ 

U – uniform deformation  $\mathbf{x} \begin{bmatrix} \ell \\ \alpha \end{bmatrix}$  – current pos.  $\mathbf{p}[\alpha]$  – sub-lat. current pos.  $\mathbf{x}[\ell]$  – unit-cell current pos.  $\mathbf{g}_i$  – current lattice basis

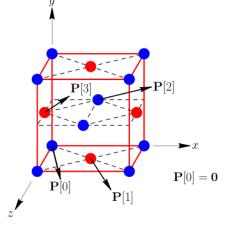




#### **CAUCHY-BORN HYPOTHESIS: ENERGY DENSITY**

• Current position vector: (Cauchy-Born kinematics,  $\alpha = 0, 1, 2, 3$ )

$$\mathbf{x} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} = \mathbf{U} \cdot \left( \mathbf{X} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} + \mathbf{P}[\alpha] \right)$$



• Atomic separation

$$r\begin{bmatrix} \ell & \ell' \\ \alpha & \alpha' \end{bmatrix} = \left\| \mathbf{x} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} - \mathbf{x} \begin{bmatrix} \ell' \\ \alpha' \end{bmatrix} \right\|$$

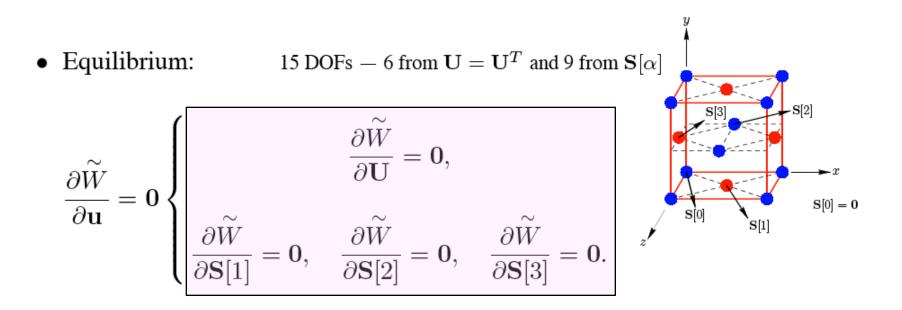
• Energy density

$$\widetilde{W}(\mathbf{U}, \mathbf{P}[1], \mathbf{P}[2], \mathbf{P}[3]; \theta) = \frac{1}{2V} \sum_{\alpha'} \sum_{\substack{[\ell] \\ \alpha'}} \phi_{\alpha\alpha'} \left( r \begin{bmatrix} \ell & 0 \\ \alpha & \alpha' \end{bmatrix}; \theta \right)$$

## PHASE TRANSFORMATIONS: ATOMISTIC



#### **CAUCHY-BORN HYPOTHESIS: EQUILIBRIUM & STABILITY**



- Stability for perturbations of  $\infty$  wavelength):
  - Cauchy-Born stability (local energy minimizer):

$$\delta \mathbf{u} \frac{\partial^2 \widetilde{W}}{\partial \mathbf{u} \partial \mathbf{u}} \delta \mathbf{u} > 0; \qquad \delta \mathbf{u} = \{ \delta \mathbf{U}, \delta \mathbf{S}[1], \delta \mathbf{S}[2], \delta \mathbf{S}[3] \}, \ \delta \mathbf{U} = \delta \mathbf{U}^T.$$





• Linearized equations of motion about equilibrium:

$$m_{\alpha} \ddot{\mathbf{u}} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} = -\sum_{\begin{bmatrix} \ell' \\ \alpha' \end{bmatrix}} \mathbf{K} \begin{bmatrix} \ell & \ell' \\ \alpha & \alpha' \end{bmatrix} \cdot \mathbf{u} \begin{bmatrix} \ell' \\ \alpha' \end{bmatrix},$$

 $m_{\alpha}$  – mass of atom  $\alpha$ ,

$$\mathbf{u} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} - \text{displacement of atom } \alpha \text{ in unit cell } \ell,$$
$$\mathbf{K} \begin{bmatrix} \ell & \ell' \\ \alpha & \alpha' \end{bmatrix} - \text{stiffness between atoms } \begin{bmatrix} \ell \\ \alpha \end{bmatrix} \text{ and } \begin{bmatrix} \ell' \\ \alpha' \end{bmatrix} \text{ cal-culated from the atomic potentials.}$$

• Initial conditions:

$$\mathbf{u} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} \Big|_{t=0} = \mathbf{u}^0 \begin{bmatrix} \ell \\ \alpha \end{bmatrix}, \quad \dot{\mathbf{u}} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} \Big|_{t=0} = \dot{\mathbf{u}}^0 \begin{bmatrix} \ell \\ \alpha \end{bmatrix}.$$

• Stability (in the sense of Lyapunov):

$$\left\|\mathbf{u}^{0}\left[\begin{smallmatrix}\ell\\\alpha\end{smallmatrix}\right]\right\|, \left\|\dot{\mathbf{u}}^{0}\left[\begin{smallmatrix}\ell\\\alpha\end{smallmatrix}\right]\right\| < \epsilon \implies \left\|\mathbf{u}\left[\begin{smallmatrix}\ell\\\alpha\end{smallmatrix}\right]\right\|, \left\|\dot{\mathbf{u}}\left[\begin{smallmatrix}\ell\\\alpha\end{smallmatrix}\right]\right\| < \delta(\epsilon).$$

# PHASE TRANSFORMATIONS: ATOMISTIC

• Phonon (normal mode) solutions: (*a*-lattice spacing, M = 4-atoms/unit-cell)

$$\mathbf{u} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} = \mathbf{\Delta} \mathbf{u}^{(q)}[\alpha] \exp \left\{ -i \left( \mathbf{k} \cdot \mathbf{X} \begin{bmatrix} \ell \\ \alpha \end{bmatrix} + \omega^{(q)}(\mathbf{k}) t \right) \right\},$$
  

$$\mathbf{k} - \text{wave number vector, } \mathbf{k} \in \left[ -\frac{\pi}{a}, \frac{\pi}{a} \right]^3;$$
  

$$q - \text{phonon index, } q = 1, 2, \dots, 3M;$$
  

$$\omega^{(q)}(\mathbf{k}) - \text{phonon frequency;}$$
  

$$\mathbf{\Delta} \mathbf{u}^{(q)}[\alpha] - \text{amplitude vector.}$$

• Phonon-stability:

$$\left(\omega^{(q)}(\mathbf{k})\right)^2 > 0$$
, for all  $\mathbf{k}$ , and all  $q$ .

NOTE: Here *a* is the interatomic distance of the cubic lattice



#### PHONON STABILITY FOR INFINITE, PERFECT LATTICE

• Block-diagonalize  $\mathbf{K}\begin{bmatrix} \ell & \ell' \\ \alpha & \alpha' \end{bmatrix}$  by taking the Fourier transform

$$m\ddot{u}(\mathbf{x},t) = -Ku(\mathbf{x},t)$$

becomes

$$\ddot{v}(\mathbf{k},t) = -\mathbb{K}v(\mathbf{k},t)$$

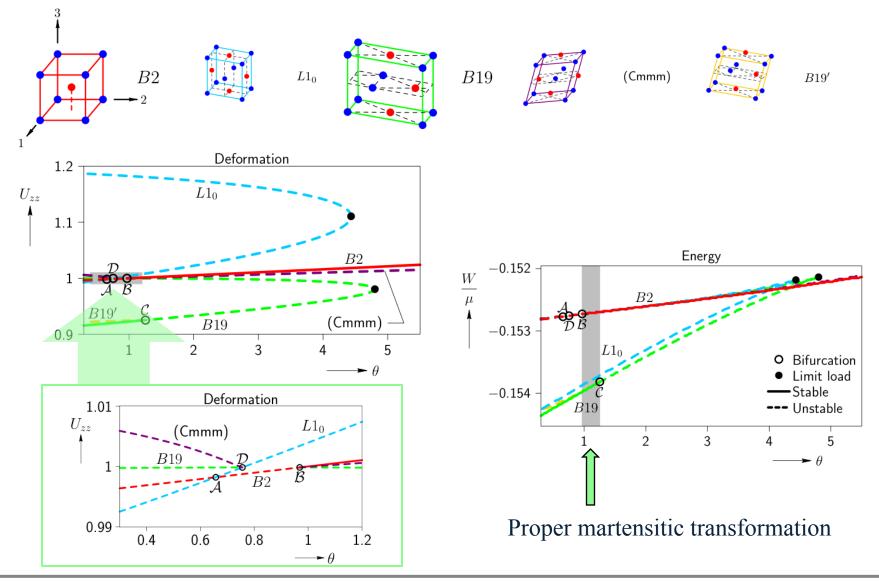
• Dynamical matrices  $-3M \times 3M$ 

$$\mathbb{K}_{p}^{j}\begin{bmatrix}\mathbf{k}\\\alpha&\alpha'\end{bmatrix} = \frac{1}{\sqrt{m_{\alpha}m_{\alpha'}}}\sum_{\ell'}G^{jk}K_{kp}\begin{bmatrix}\mathbf{0}&\ell'\\\alpha&\alpha'\end{bmatrix}\exp\left\{-i\mathbf{k}\cdot\left(\mathbf{X}\begin{bmatrix}\ell'\\\alpha'\end{bmatrix} - \mathbf{X}\begin{bmatrix}\mathbf{0}\\\alpha\end{bmatrix}\right)\right\}.$$

- The  $(\omega^{(q)}(\mathbf{k}))^2$ , are the 3M eigen-values of  $\mathbb{K}_p^j \begin{bmatrix} \mathbf{k} \\ \alpha & \alpha' \end{bmatrix}$  for each  $\mathbf{k}$ .
- Numerically use a uniform grid in k domain  $([-\frac{\pi}{a}, \frac{\pi}{a})^3)$



#### **CAUCHY-BORN & PHONON STABILITY RESULTS**



PHASE TRANSFORMATIONS: ATOMISTIC



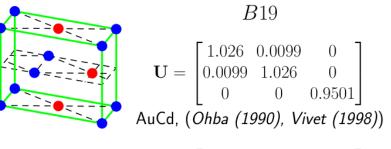
#### **COMPARISON OF CALCULATIONS WITH EXPERIMENTS**

 $B2 \Longrightarrow B19$ 

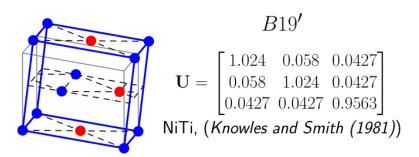
Martensitic Transformation

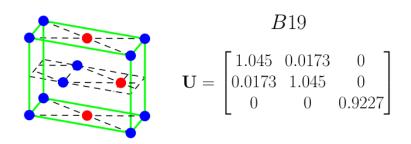
• Experimental right stretch tensor

• Simulated right stretch tensor ( $\theta = 1.0$ )



$$\mathbf{U} = \begin{bmatrix} 1.042 & 0.0195 & 0\\ 0.0195 & 1.042 & 0\\ 0 & 0 & 0.9178 \end{bmatrix}$$
CuAlNi, (*Bhatacharya, James (1999*))





- A 4-atom Cauchy-Born lattice model finds a B19 phase with a stable segment which overlaps with the stable segment of the B2 phase.
- Results indicate a hysteretic, proper martensitic transformation in this biatomic lattice model.
- Reasonable agreement with experiments.



#### WHAT HAVE LEARNED FROM THIS APPLICATION

- Usefulness of LSK aymptotics to a) guide the nonlinear calculations near critical points and b) serve as a check of the numerical calculations.
- Important concept of scale. There are global (long wavelength) and local (of the order of unit cell dimensions) instabilities. One has to check for both!
- Important concept of phonon spectra (same as Bloch wave representation but for discrete systems) can check stability of an infinite periodic system by analyzing only one unit cell.
- Analytical methods are indispensable for effective and intelligent numerical calculations
- Work presented explains only thermal behavior of SMA's. Stress influence on SMA's more complicated, and has already been studied using same ideas.
- Ultimate goal: understand reasons for phase transformations and design materials at the atomic scale.